

# Rapid electroplating of photocatalytically highly active TiO<sub>2</sub>-Zn nanocomposite films on steel

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Nanocomposite films consisting of TiO<sub>2</sub> and Zn with thickness of 10–15 μm (TiO<sub>2</sub>-Zn) have been electrodeposited on steel plates by rapid plating from a ZnSO<sub>4</sub>-based bath ( $I_d > 10 \text{ A dm}^{-2}$ ). Upon addition of NH<sub>4</sub>NO<sub>3</sub> to the bath (<0.3 g L<sup>-1</sup>), the uptake of TiO<sub>2</sub> in the film significantly increased. Glow discharge optical emission spectrometry clarified that TiO<sub>2</sub> particles were incorporated throughout the film and the loaded amount increased near the surface. The first-order rate constant ( $k/h^{-1}$ ) for gas-phase CH<sub>3</sub>CHO oxidation was employed as an indicator of the photocatalytic activity. The  $k$  value for the TiO<sub>2</sub>-Zn film prepared at  $I_d = 12 \text{ A dm}^{-2}$  ( $0.20 \text{ h}^{-1}$ ) was comparable to that for the sample from a ZnCl<sub>2</sub>-based bath at  $I_d = 4 \text{ A dm}^{-2}$  ( $0.27 \text{ h}^{-1}$ ). X-ray diffraction measurements indicated that a TiO<sub>2</sub>-ZnO nanocomposite layer was generated on the surface by the heat treatment in air at 673 K for 6 h. Consequently, the photocatalytic activity was further improved ( $k = 0.29 \text{ h}^{-1}$ ); this effect was explained in terms of the synergy of TiO<sub>2</sub> and ZnO in photocatalysis. © 2001 Kluwer Academic Publishers

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## 1. Introduction

Nanocomposite films having both the positive attributes of semiconductors in colloidal (large surface area) and thin-film (usability without filtration) forms are expected to be suitable for use as a photocatalyst. The research group of Tacconi and Rajeshwar reported a method of electrodepositing a nanocomposite film consisting of TiO<sub>2</sub> particles and Ni (TiO<sub>2</sub>-Ni) on polycrystalline Au electrodes [1, 2]. On the other hand,

steel, an essential material in buildings, attracts much interest as a substrate of photocatalytic films competent to purify polluted air and water. Electrodeposition of Zn films on steel is widely practiced for corrosion resistance. We have recently prepared a TiO<sub>2</sub>-Zn nanocomposite film on steel from a ZnCl<sub>2</sub>-based bath at a fairly small current density ( $I_d$ ) of  $4 \text{ A dm}^{-2}$  [3]. A high photocatalytic activity was revealed for the sample; however, a more rapid process applicable

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to on-line treatment is required from the practical viewpoint.

This paper describes rapid electroplating of TiO<sub>2</sub>-Zn nanocomposite films on steel plates from a ZnSO<sub>4</sub>-based bath ( $I_d > 10 \text{ A dm}^{-2}$ ). In view of the photocatalytic activity, the optimum conditions for preparing the film are explored.

## 2. Experimental procedure

### 2.1. Electroplating of TiO<sub>2</sub>-Zn

#### nanocomposite films on steel

Steel with a size of 2.5 cm × 5.0 cm and 0.08 cm in thickness (Nisshin Steel Co. Ltd.) was used as a substrate. Prior to the electrodeposition, steel was washed successively with acetone, a neutral detergent, water and an acidic water (HCl : H<sub>2</sub>O = 1 : 1 v/v). The electrodeposition bath contained 350 g L<sup>-1</sup> ZnSO<sub>4</sub>, 30 g L<sup>-1</sup> (NH<sub>4</sub>)<sub>2</sub>SO<sub>4</sub> and TiO<sub>2</sub> particles (average diameter = 6 nm, BET surface area = 260 m<sup>2</sup> g<sup>-1</sup>, anatase, AMT-100, Tayca Co.); the pH of the bath was adjusted at 4. The concentration of NH<sub>4</sub>NO<sub>3</sub> (x/g L<sup>-1</sup>) and the weight of TiO<sub>2</sub> particles (y/g) were varied at  $0 \leq x \leq 0.75$  and  $0 \leq y \leq 200$ , respectively. Zn (or TiO<sub>2</sub>-Zn) films were electrodeposited on steel by flowing a constant current of  $I_d$  ( $4 \leq I_d \leq 20 \text{ A dm}^{-2}$ ) between the steel cathode and a Zn anode at 45°C. The area of the steel electrodeposited and the quantity of electricity were fixed at 25 cm<sup>2</sup> and 720 C, respectively. The nanocomposite film prepared from the bath containing NH<sub>4</sub>NO<sub>3</sub> and TiO<sub>2</sub> with respective concentration of  $x$  and  $y$  was denoted as TiO<sub>2</sub>-Zn( $x$ ,  $y$ )/steel. Post-annealing was carried out in air at 673 K for 2–6 h in order to oxidize the Zn matrix.

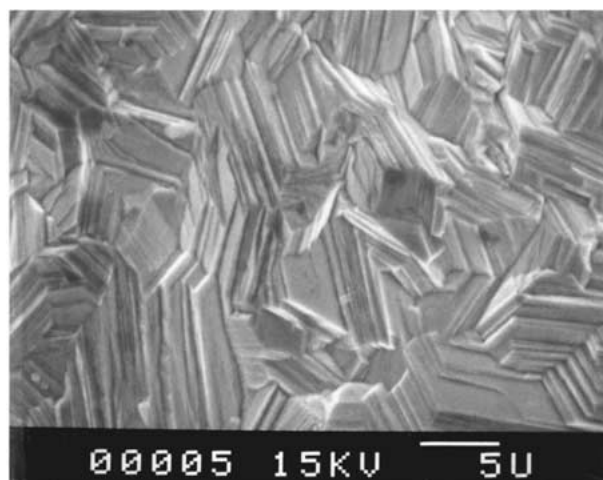
### 2.2. Characterization of the TiO<sub>2</sub>-Zn nanocomposite films

The Zn in the TiO<sub>2</sub>-Zn film (12.5 cm<sup>2</sup>) was firstly dissolved into a HCl aq. solution (20 mL, conc. HCl: H<sub>2</sub>O = 1:1 v/v), and then 10 mL of a H<sub>2</sub>SO<sub>4</sub> aq. solution (conc. H<sub>2</sub>SO<sub>4</sub> : H<sub>2</sub>O = 1 : 1 v/v) was added and heated to dissolve the TiO<sub>2</sub> particles completely. After the homogeneous solution had been further diluted to 50 mL with H<sub>2</sub>O, it was analyzed by induced coupled plasma spectroscopy (ICPS-1000, Shimadzu) to determine the amount of the TiO<sub>2</sub> uptake.

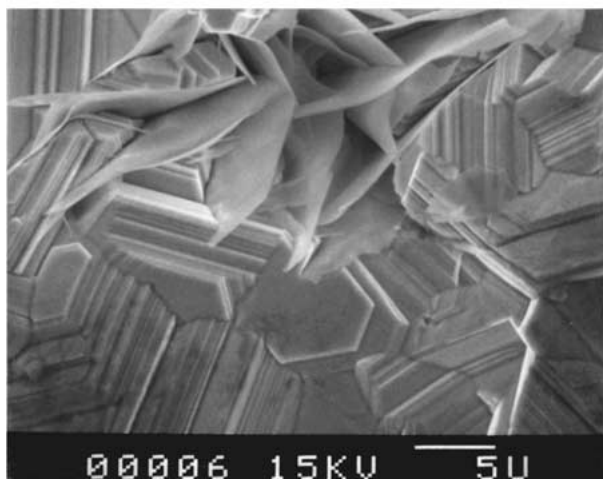
X-ray diffraction (XRD) patterns of TiO<sub>2</sub>-Zn/steel samples were obtained on a Rigaku RINT 2500. The measurements were performed in the range of 5°–105° at 8° min<sup>-1</sup> using Cu-K<sub>α</sub> as a x-ray source (40 kV-80 mA).

Diffuse reflectance spectra of the samples were measured using a Hitachi U-400 spectrophotometer in the wavelength range between 300 and 500 nm. The reflectance was recorded with respect to a reference of Al<sub>2</sub>O<sub>3</sub>.

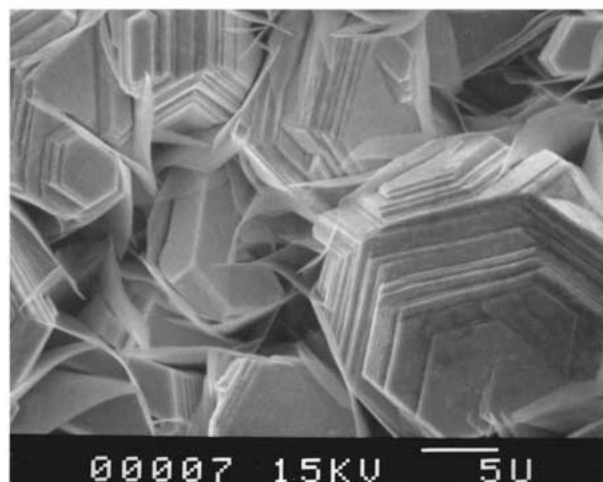
Depth profiles of the TiO<sub>2</sub>-Zn film compositions were determined using a glow discharge optical emission spectrometer (GDOES, Rigaku GDS-3860). The measurements were carried out at a power of 30 W



(A)



(B)



(C)

Figure 1 SEMs of Zn/steel samples prepared at  $I_d = 20 \text{ A dm}^{-2}$ : A,  $x = 0$ ; B,  $x = 0.2$ ; C,  $x = 0.3$ .

under an Ar flow (180 mL min<sup>-1</sup>). The inner diameter of the anode was 4 mm.

### 2.3. Evaluation of photocatalytic activity

The photocatalytic activities of TiO<sub>2</sub>-Zn/steel samples in the oxidation of CH<sub>3</sub>CHO were measured [4, 5].

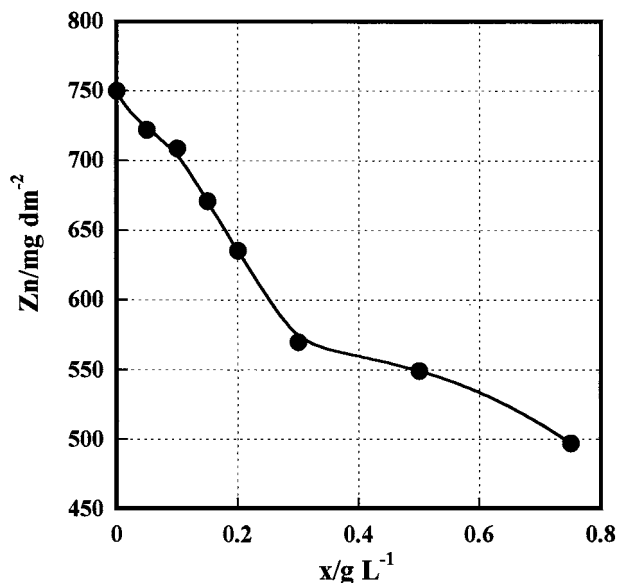


Figure 2 Plots of the weight of Zn vs.  $x$ : electricity = 720 C.

A 500-ppm standard  $\text{CH}_3\text{CHO}$  gas ( $\text{CH}_3\text{CHO}/\text{N}_2$ ) was introduced into a reaction chamber (0.64 L), and diluted with air so that its initial concentration was controlled within the  $200 \pm 50$  ppm range. After the adsorption equilibrium of  $\text{CH}_3\text{CHO}$  had been achieved in the dark, front-face irradiation ( $\lambda > 300$  nm,  $I_{320-400} = 2.9 \text{ mW cm}^{-2}$ ) of the sample (apparent surface area =  $19.6 \text{ cm}^2$ ) was carried out using a 300 W Xe lamp (Wacom, model XDS-301S) at room temperature. The concentration of  $\text{CH}_3\text{CHO}$  was determined by gas chromatography (Shimadzu GC-14B).

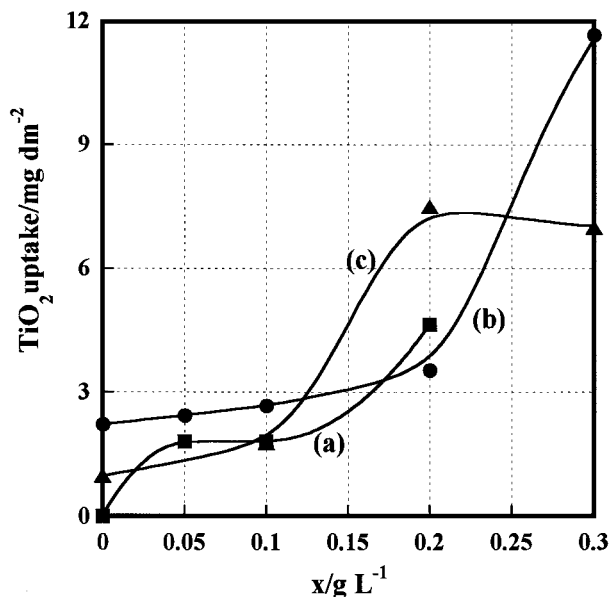


Figure 4 Dependence of the loaded amounts of  $\text{TiO}_2$  on  $x$ : a,  $I_d = 8 \text{ A dm}^{-2}$ ; b,  $I_d = 12 \text{ A dm}^{-2}$ ; c,  $I_d = 20 \text{ A dm}^{-2}$ .  $y$  was fixed at 100.

### 3. Results and discussion

#### 3.1. $\text{NH}_4\text{NO}_3$ additive effect on the Zn electroplating

The effect of  $\text{NH}_4\text{NO}_3$  addition to the bath on the Zn electroplating was examined with its amount varied. Fig. 1 shows the scanning electron micrographs (SEMs) of Zn/steel samples prepared at  $I_d = 20 \text{ A dm}^{-2}$ : A,  $x = 0$ ; B,  $x = 0.2$ ; C,  $x = 0.3$ . Sample A is a dense polycrystalline film consisting of many hexagonal platelets. Sample B has holes surrounded by thin leaves, and the number of the holes increases

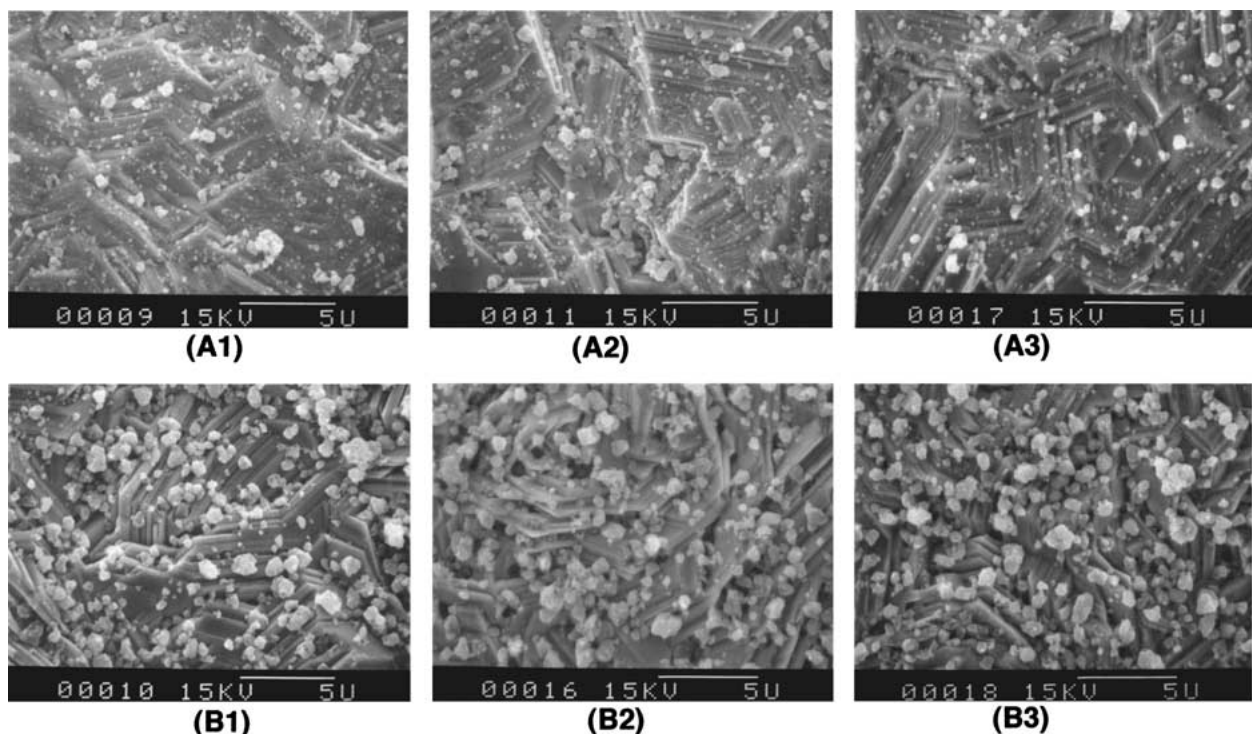
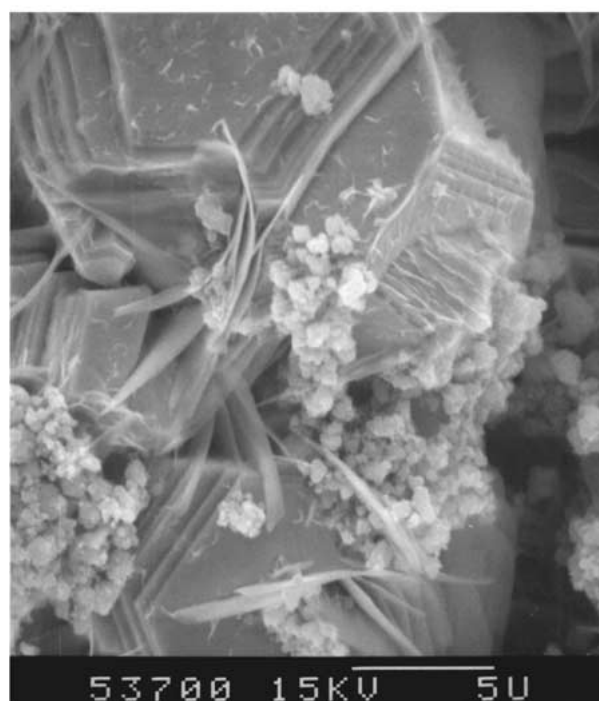


Figure 3 A, SEMs of  $\text{TiO}_2\text{-Zn}(0,100)/\text{steel}$  samples: A1,  $I_d = 8 \text{ A dm}^{-2}$ ; A2,  $I_d = 12 \text{ A dm}^{-2}$ ; A3,  $I_d = 20 \text{ A dm}^{-2}$ . B, SEMs of  $\text{TiO}_2\text{-Zn}(0.3,100)/\text{steel}$  samples: B1,  $I_d = 8 \text{ A dm}^{-2}$ ; B2,  $I_d = 12 \text{ A dm}^{-2}$ ; B3,  $I_d = 20 \text{ A dm}^{-2}$ .



(A)

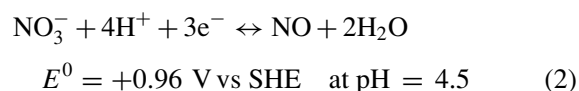
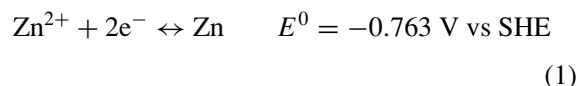


(B)

Figure 5 Surface SEM images for the TiO<sub>2</sub>-Zn nanocomposite films prepared at  $I_d = 8 \text{ A dm}^{-2}$ : A, TiO<sub>2</sub>-Zn(0.1,10)/steel; B, TiO<sub>2</sub>/Zn(0.5,10)/steel.

in sample C. A similar trend was observed for samples formed at  $I_d = 8$  and  $12 \text{ A dm}^{-2}$ . Fig. 2 shows the weight of Zn electrodeposited at constant electricity of  $720 \text{ C}$  as a function of  $x$  ( $0 \leq x \leq 0.75$ ). The amount of Zn decreases monotonically with an increase in  $x$ . The electroplating efficiencies ( $\eta$ ) for  $x = 0$  and  $x = 0.3$  were calculated to be 77% and 58%, respectively, by using these data. Since some redox reactions of  $\text{NO}_3^-$  ions have more positive standard electrode

potentials ( $E^0$ ) than that for the redox couple of  $\text{Zn}^{2+}/\text{Zn}$  (Equation 1) [6], the decrease in  $\eta$  may result from the side reactions involving one expressed by Equation 2.



### 3.2. Additive effect of $\text{NH}_4\text{NO}_3$ on the TiO<sub>2</sub>-Zn nanocomposite electroplating

Fig. 3A shows the SEMs of TiO<sub>2</sub>-Zn(0,100)/steel samples: A1,  $I_d = 8 \text{ A dm}^{-2}$ ; A2,  $I_d = 12 \text{ A dm}^{-2}$ ; A3,  $I_d = 20 \text{ A dm}^{-2}$ . In every sample, a number of TiO<sub>2</sub> particles smaller than ca.  $1 \mu\text{m}$  are dispersed on the surface of the Zn film. Fig. 3B are the SEMs of TiO<sub>2</sub>-Zn(0.3,100)/steel samples: B1,  $I_d = 8 \text{ A dm}^{-2}$ ; B2,  $I_d = 12 \text{ A dm}^{-2}$ ; B3,  $I_d = 20 \text{ A dm}^{-2}$ . The addition of  $\text{NH}_4\text{NO}_3$  significantly increases both the average size and the uptake of TiO<sub>2</sub> aggregates. Further, the amount of TiO<sub>2</sub> incorporated into the film was determined by ICPS. Fig. 4 shows the variation of the TiO<sub>2</sub> amount as a function of  $x$  ( $y = 100$ ): a,  $I_d = 8 \text{ A dm}^{-2}$ ; b,  $I_d = 12 \text{ A dm}^{-2}$ ; c,  $I_d = 20 \text{ A dm}^{-2}$ . Irrespective of current density, the amount of TiO<sub>2</sub> increases with increasing  $x$  at  $0 \leq x \leq 0.3$ . A maximum amount of TiO<sub>2</sub> incorporated from the  $\text{ZnSO}_4$ -based bath is obtained under conditions of  $I_d = 12 \text{ A dm}^{-2}$ ,  $x = 0.3$  and  $y = 100$  ( $11.8 \text{ mg dm}^{-2}$ ), which is smaller than that from the  $\text{ZnCl}_2$ -based bath by a factor of ca. 3 [7]. Fig. 5 shows the surface SEM images for the TiO<sub>2</sub>-Zn nanocomposite films prepared at  $I_d = 8 \text{ A dm}^{-2}$ : A, TiO<sub>2</sub>-Zn(0.1,10)/steel; B, TiO<sub>2</sub>/Zn(0.5,10)/steel. Apparently, TiO<sub>2</sub> particles are preferentially incorporated into the holes generated by the  $\text{NH}_4\text{NO}_3$  addition. This seems to be the main reason for the increase in the loaded amount of TiO<sub>2</sub> with the addition of  $\text{NH}_4\text{NO}_3$ . Fig. 6 shows the GDOES depth profiles for the TiO<sub>2</sub>-Zn nanocomposite films prepared at  $I_d = 12 \text{ A dm}^{-2}$ : A, TiO<sub>2</sub>-Zn(0.3,100)/steel; B, TiO<sub>2</sub>-Zn(0,100)/steel. Depth profile B demonstrates that a small amount of TiO<sub>2</sub> particles is present only near the surface of the film. In sample A, Ti is distributed throughout the film, and its signal intensity remarkably increases as it approaches the surface.

### 3.3. Heat treatment of TiO<sub>2</sub>-Zn nanocomposite films

The crystallinity change of the TiO<sub>2</sub>-Zn nanocomposite film with heat treatment was examined. Fig. 7 shows the XRD patterns for the films: a, TiO<sub>2</sub>-Zn(0.3,100)/steel; b, TiO<sub>2</sub>-Zn(0.3,100)/steel heated at  $673 \text{ K}$  for 6 h; c, Zn(0.3,0)/steel heated at  $673 \text{ K}$  for 6 h. In sample (a), a diffraction peak from the (101) plane of anatase is observed at  $2\theta = 25.3^\circ$  together with several peaks due to Zn crystals. In pattern (b), all the peaks of Zn

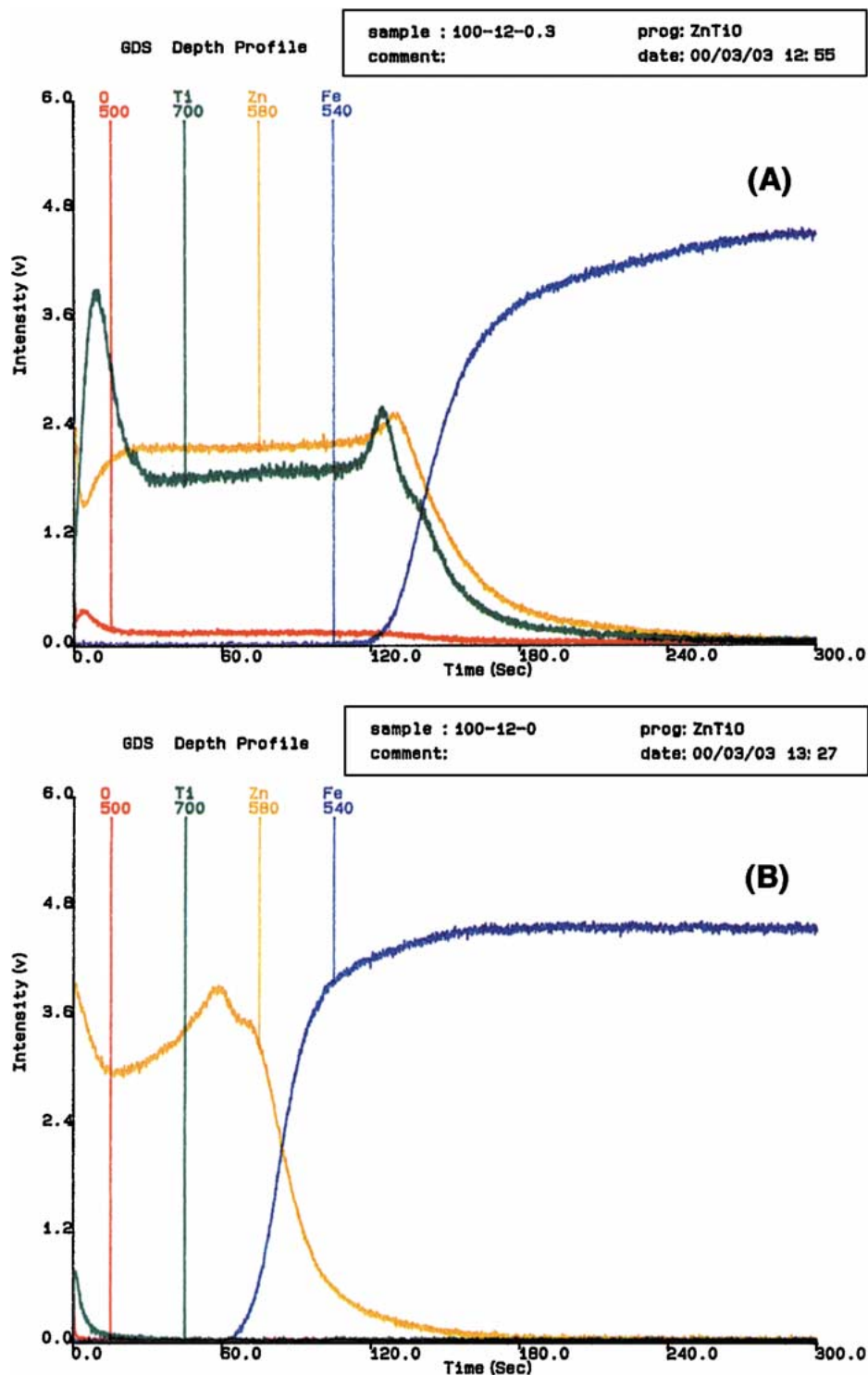


Figure 6 GDOES depth profiles for the TiO<sub>2</sub>-Zn nanocomposite films prepared at  $I_d = 12 \text{ A dm}^{-2}$ : A, TiO<sub>2</sub>-Zn(0.3,100)/steel; B, TiO<sub>2</sub>-Zn(0,100)/steel.

disappear, and new peaks assignable to those of hexagonal ZnO crystals appear [8]. The latter peak positions are in complete agreement with those in pattern (c). These findings indicate that the surface of sample (b) consists of a nanocomposite of TiO<sub>2</sub> and ZnO (TiO<sub>2</sub>-Zn( $x$ ,  $y$ )/steel). Fig. 8 shows diffuse reflectance spectra of the following samples: a, Zn(0.3,0)/steel; b, TiO<sub>2</sub>-Zn(0.3,100)/steel; c, ZnO(0.3,0)/steel; d, TiO<sub>2</sub>-ZnO(0.3100)/steel. Spectrum (a) indicates high reflectivity (>60%) characteristic of metals in the

300–500 nm range. In spectrum (b), the band gap transition of anatase TiO<sub>2</sub> (band gap,  $E_g \cong 3.2 \text{ eV}$ ) is responsible for the decrease in reflectivity at  $\lambda < 385 \text{ nm}$  [9]. Also, the decrease of reflectivity at  $\lambda < 400 \text{ nm}$  in spectrum (c) results from the band gap absorption of ZnO crystals (Fig. 7c). Since the  $E_g$  of ZnO is equal to that of TiO<sub>2</sub> [9], the redshift of the absorption edge by 15 nm (400 nm–385 nm) may be induced by oxygen vacancy of ZnO [10]. The significant decrease in reflectivity at  $\lambda < 420 \text{ nm}$  in spectrum (d) is ascribable

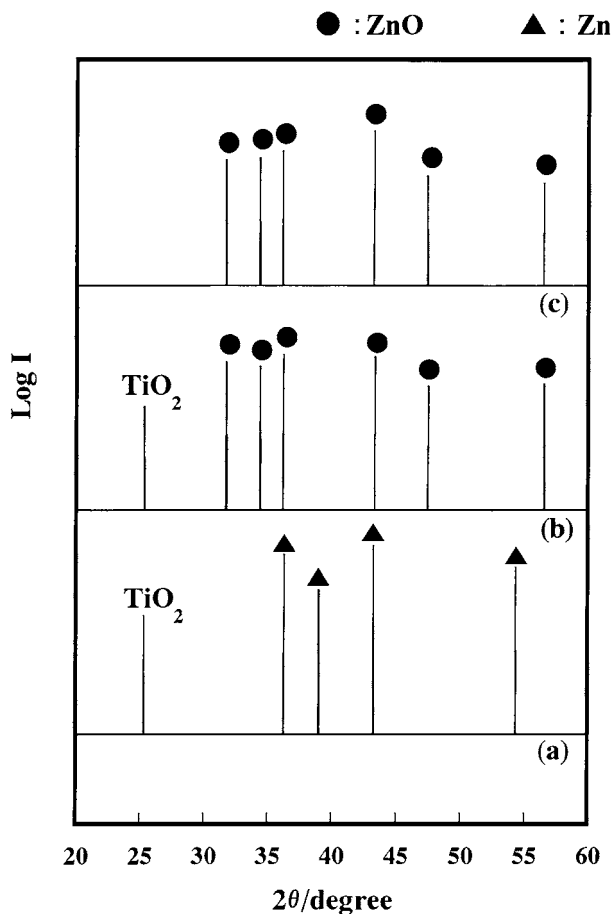


Figure 7 XRD patterns for the films: a, TiO<sub>2</sub>-Zn(0.3,100)/steel; b, TiO<sub>2</sub>-Zn(0.3,100)/steel heated at 673 K for 6 h; c, Zn(0.3,0)/steel heated at 673 K for 6 h.

to the overlap of the absorption of TiO<sub>2</sub> and ZnO. The further redshift of the absorption edge by 20 nm (420 nm–400 nm) suggests that the high population of the TiO<sub>2</sub> particles on the surface (Fig. 6A) suppresses the oxidation of Zn surfaces. This in turn explains the high reflectivity at  $\lambda > 420$  nm.

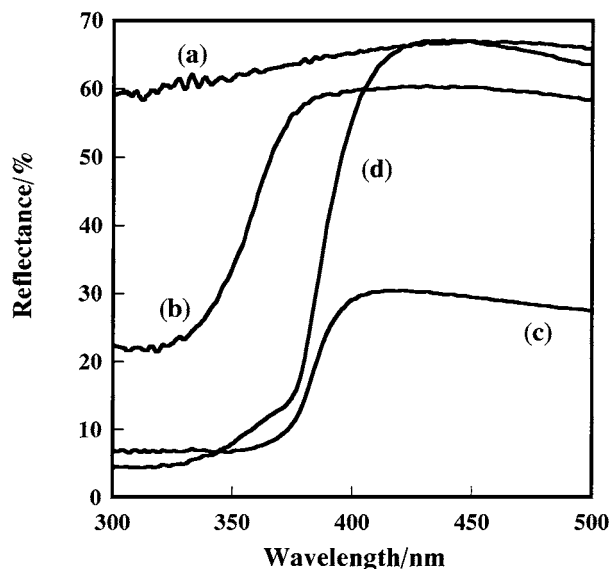


Figure 8 Diffuse reflectance spectra of the following samples: a, Zn(0.3,0)/steel; b, TiO<sub>2</sub>-Zn(0.3,100)/steel; c, ZnO(0.3,0)/steel; d, TiO<sub>2</sub>-ZnO(0.3100)/steel.

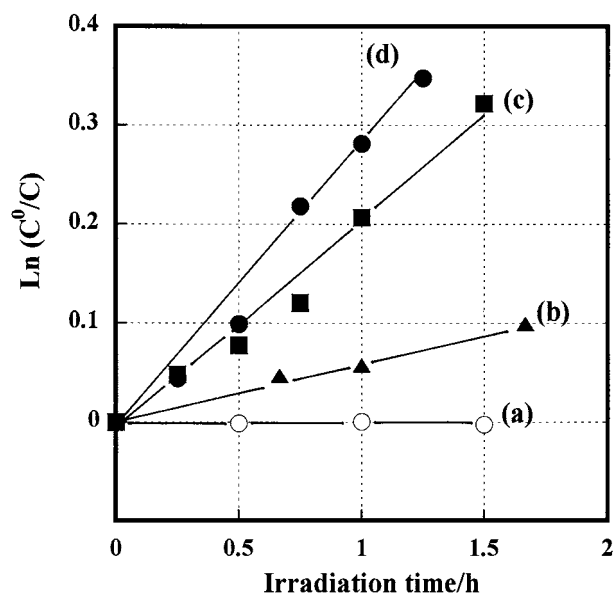


Figure 9 Photocatalytic activities of several samples produced at  $I_d = 12 \text{ A dm}^{-2}$  for CH<sub>3</sub>CHO decomposition: (a), Zn(0.3,0)/steel; (b), ZnO(0.3,0)/steel; (c), TiO<sub>2</sub>-Zn(0.3,100)/steel; (d), TiO<sub>2</sub>-ZnO(0.3,100)/steel.  $C^0$  and  $C$  denote the concentrations of CH<sub>3</sub>CHO at  $t = 0$  and  $t = t$ , respectively.

### 3.4. Photocatalytic activity of TiO<sub>2</sub>-Zn nanocomposite films

Fig. 9 shows the photocatalytic activities of several samples produced at  $I_d = 12 \text{ A dm}^{-2}$  for CH<sub>3</sub>CHO decomposition: (a), Zn(0.3,0)/steel; (b), ZnO(0.3,0)/steel; (c), TiO<sub>2</sub>-Zn(0.3,100)/steel; (d), TiO<sub>2</sub>-ZnO(0.3,100)/steel.  $C^0$  and  $C$  denote the concentrations of CH<sub>3</sub>CHO at  $t = 0$  and  $t = t$ , respectively. In every system, the plots yields a straight line with a slope giving an apparent rate constant ( $k$ ). No decomposition of CH<sub>3</sub>CHO occurs in system (a). The fact that both TiO<sub>2</sub> (and/or ZnO) and illumination ( $\lambda > 300$  nm) are required for the CH<sub>3</sub>CHO decomposition suggests that this reaction is induced by the band gap excitation of TiO<sub>2</sub> (and/or ZnO). In system (b), the concentration of CH<sub>3</sub>CHO decreases with illumination ( $k = 0.06 \text{ h}^{-1}$ ); this can be attributed to the photocatalytic activity of ZnO generated as the result of surface oxidation. It is known that ZnO has a photocatalytic activity for the oxidation of CH<sub>3</sub>CHO [11]. Sample (c) has a considerably higher activity due to TiO<sub>2</sub> particles incorporated into the film ( $k = 0.20 \text{ h}^{-1}$ ). Noticeably, the activity of sample (d) ( $k = 0.29 \text{ h}^{-1}$ ) is further greater than that of sample (c). The sum of the  $k$  values of samples (b) and (c) is almost equal to the  $k$  value of sample (d). This finding suggests that ZnO as well as TiO<sub>2</sub> acts as a photocatalyst in sample (d), whereas Zn is only a matrix for fixing TiO<sub>2</sub> particles in sample (c).

### 4. Conclusions

TiO<sub>2</sub>-Zn nanocomposite films were successfully formed on steel plates by rapid plating from a ZnSO<sub>4</sub>-based bath ( $I_d > 10 \text{ A dm}^{-2}$ ). The addition of NH<sub>4</sub>NO<sub>3</sub> to the bath significantly increased the uptake of TiO<sub>2</sub> in the film. The GDOES analyses clarified that TiO<sub>2</sub> particles are incorporated throughout the film and its

amount increases near the surface. The resultant film exhibited a photocatalytic activity for CH<sub>3</sub>CHO oxidation comparable to that of the sample prepared from a ZnCl<sub>2</sub>-based bath at  $I_d = 4 \text{ A dm}^{-2}$ . Heat treatment at 673 K for 6 h generated a TiO<sub>2</sub>-ZnO nanocomposite layer on the surface, improving the photocatalytic activity further. This effect was explained in terms of the synergy of TiO<sub>2</sub> and ZnO in photocatalysis. This research has presented a feasible method for endowing a high photocatalytic activity with steel plates.

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